

Research Article

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Silica and kaolin reinforced aluminum matrix composite for heat storage

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Abstract: This study used aluminum scraps to produce a secondary aluminum metal matrix for heat storage analyses. Silica and kaolin reinforced aluminum metal matrix composites were successfully produced *via* stir casting. The X-ray diffraction (XRD) and scanning electron microscopy (SEM) were employed for phase and microstructure characterization. XRD revealed alumina (corundum), aluminum and kyanite phases while SEM indicated pores in the composites. Density, average specific heat (from 30 to 200°C), thermal conductivity, and hardness tests were carried out. Total heat energy stored per kg, from 30 to 200°C, was obtained. The inclusion of 7.5 and 15 wt% kaolin increased the specific heat of the matrix from 474.3 to 564.57 J·kg⁻¹·°C⁻¹ and 474.3 to 679.03 J·kg⁻¹·°C⁻¹, respectively. Likewise, adding 7.5 and 15 wt% silica sand increased the thermal conductivity of the matrix from 154.99 to 175.62 W·m⁻¹·°C⁻¹ and 154.99 W·m⁻¹·°C⁻¹ to 181.38 W·m⁻¹·°C⁻¹, respectively. The addition of 7.5 wt% silica sand and 7.5 and 15 wt% kaolin increased the hardness value of the matrix from 72.11 to 73.11 HB, 72.11 to 81.38 HB, and 72.11 to 82 HB, respectively. Hardness of the composites reinforced with kaolin is higher than that of the composites reinforced with silica sand. This is attributed to the higher molecular weight of kaolin. Significant

increase in specific heat and thermal conductivity was achieved.

Keywords: kaolin, microstructure, secondary aluminum, silica sand, specific heat

1 Introduction

Recent studies have shown that automobile and aircraft parts are often disposed of habitually because they occupy a large amount of land resources and pose a severe ecological environmental threat [1]. Secondary (recycled) aluminum exists in the form of alloys compared with those obtained from industrial primary processes. Aluminum production *via* the primary process is usually done by electrolytic reduction, which involves very high energy consumption imposed by high current [2]. However, scrap aluminum alloy has a lower melting temperature, shorter melting time, lower metal burning loss, and less energy consumption which is much more economical in terms of production costs [2,3]. Apart from cost reduction, it also reduces the amount of CO₂ emission, thereby lowering the global warming potential of mining pure aluminum [4]. Thus, it is essential to take advantage of cost reduction associated with scrap aluminum materials and improve its limiting properties.

In recent times, significant research and development is geared towards wrought aluminum alloys by using scrap aluminum recycling technology and equipment [4–6]. Meanwhile, the use of scrap aluminum alloy wheels as a matrix material for the fabrication of aluminum metal matrix composite has also gained significant research interest [7–11]. Secondary or recycled aluminum metal matrix composite has emerged as the material of choice for application in thermal management, aerospace, recreational, and infrastructure industries [12]. This is due to its high structural efficiency, excellent wear resistance, and attractive thermal and electrical characteristics [4,13–16].

Several research reports have been published on the fabrication of aluminum metal matrix composites using different approaches such as stir casting [11,13,17,18],

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squeeze casting [13,17,18], friction stir casting [19,20], and powder metallurgy [16]. Among these techniques, the stir casting process has been widely used to fabricate aluminum metal matrix composites due to its low-cost implication [21]. However, components produced from the stir casting method suffer from high porosity, thus, limiting their use for high strength components [18–21]. However, this limiting property of aluminum metal matrix composite can be improved by incorporating suitable reinforcement materials into the matrixes using appropriate technologies [22–24]. For instance, Farajollahi *et al.* [25] studied the effect of adding Ni on the microstructure and mechanical properties of aluminum alloy using the stir casting method at the speed of 300 rpm for 20 min at 750°C. For up to 4 wt% of nickel added, the significant increase in the hardness (from about 92 to 144.05 VHN) and strength of the as-cast aluminum alloy used was because of the formation of intermetallic compounds. Whereas, beyond 3 wt%, *i.e.*, addition of up to 4.5 wt% of nickel resulted in Al_6CuMg_4 precipitates, which increased the porosity and reduced the hardness of the composites to 118.54 VHN.

Similarly, *via* the stir casting method, Shayan *et al.* [26] used TiO_2 , Al, and Cu powder mixture as reinforcement in an aluminum alloy matrix to test for the resulting mechanical properties of the fabricated composites. The stirring was carried out at 750°C for 15 min at a stirring speed of 500 rpm. As the quantity of reinforcements increased, they recorded an increase in hardness and ultimate tensile strength. Kumar *et al.* [27] reinforced aluminum matrix with 4, 8, and 12% Si_3N_4 by stir casting at 200 rpm. After treating the composites with heat, they observed a reduction in the porosity of the composites. Using stir casting at 800°C, Reddy *et al.* [28] varied the composition of SiC and B_4C in an aluminum matrix and observed the effects on the mechanical properties of the composite. An improvement in the mechanical properties was reported with a higher concentration of reinforcements relative to the composites with a lower concentration of reinforcements, which then gave an increase in tensile strength and hardness of 107.64–128.24 MPa and 30.43–45.8 BHN, respectively.

In a study conducted by Shirvanimoghaddam *et al.* [29], the volume fraction of B_4C , fabrication temperature, stirring time, and particle size were varied against the mechanical and physical properties of aluminum used as matrix. According to the results, an increase in the tensile strength and hardness were recorded. The wettability, agglomeration, and dispersion of the particles were affected by the temperature, stirring speed, and time of stirring. In a similar study by Selvakumar *et al.* [30], the tensile strength of the material increased from 222 to 303 MPa when 0, 6, 12, and 18 vol% of molybdenum particles were added to the aluminum matrix.

Using the stir casting technique, Shuvho *et al.* [31] investigated the effect of reinforcing aluminum matrix with Al_2O_3 , SiC, and TiO_2 of average particle size (65 microns) on the mechanical properties of the resulting composites. The stirring process was carried out at about 660°C at 400 rpm for 8 min. The increase in the quantity of the SiC particles led to a corresponding increase in hardness, tensile and yield strengths of the matrix from 29.7 to 98.56 HB, 110.07 to 148.62 $N\cdot mm^{-2}$, and 78.6 to 109.8 $N\cdot mm^{-2}$, respectively.

Using stir casting technique while operating at 500 rpm and 750°C for 10 min, Daniel-Mkpume *et al.* [32] reinforced aluminum matrix with SiO_2 particles of 53 microns average particle size in weight fractions of 3, 7, 11, 15, and 19 wt%. They observed a decrease in the density as the weight fraction of SiO_2 increased. They also observed an increase in the ultimate tensile strength from 49.73 to 65.64 MPa. Mallikarjuna and Basavaraj [5] investigated the effects of varying SiO_2 particle sizes (106, 150, 250, and 335 microns) and varying SiO_2 weight fractions from 3 to 12 wt% on the mechanical properties of the resulting composites. However, no significant effect on the mechanical properties was observed due to the different particle sizes. Altinkok and Koker [6] studied the effect of reinforcing aluminum matrix with Al_2O_3/SiC on the resulting composite's bending strength and hardness resistance. For 10 vol% of Al_2O_3/SiC , the bending strength and hardness resistance reduced from 481.2 to 378.3 MPa and 118 to 77 HB, respectively, as the SiC particle size increased. Pramod *et al.* [33] used 150 microns of Al_2O_3 to reinforce aluminum from 2 to 6% weight fraction by stir casting at 710°C for 10 min at a stirring speed of 400 rpm. This was done to observe the effect on density and wear height loss within an applied load and sliding distance. Judging from their results, the composite containing 6 wt% of reinforcement had the highest performance against wear resistance. Gupta *et al.* [34] investigated the effect of reinforcing pure aluminum and eutectic aluminum alloy with silica particles up to 25 wt% through stir casting on the mechanical properties of the composites. They observed a reaction between aluminum and silica which gave alumina and silicon as products at 723 K. Adding up to 20 wt% silica to the pure aluminum increased the hardness from 52 to 78 BHN but decreased the ultimate tensile strength from 92 to 62 MPa.

On the other hand, adding 20 wt% silica to the eutectic alloy did not affect the hardness but decreased the ultimate tensile stress from 184 to 112 MPa. Using the stir casting method, Hemanth [35] added 3, 6, 9, and 12 wt% quartz particles to aluminum alloy by stirring at 700 rpm at a temperature of 750°C in a bid to investigate their effects on the mechanical properties. Observation of uniform distribution of particulates from the microstructural analysis

was made. The hardness, ultimate tensile strength, and Young's modulus increased from 80 to 125 HV, 172 to 257 MPa, and 68 to 89 GPa, respectively for up to 9 vol% of quartz-reinforcements. The thermal conductivities of the composites reduced as the particles increased. Singh *et al.* [36] worked on varying the amount of silica in an aluminum alloy matrix and studied the effects on the mechanical properties of the resulting composites. They used the stir casting method with a stirring speed of 330 rpm at a temperature of 850 for 5 min and observed an increase in hardness (from 87 to 128.66 HV), tensile strength, and impact energy with the increase in reinforcement.

Although several articles have reported on the fabrication of aluminum metal matrix composites by stir casting method, only few reports are available on the thermal properties. Despite the large volume of literature available on the use of reinforcements in aluminum matrix for energy applications, no work has been done on the use of kaolin as reinforcement in relation to its application for thermal energy storage to the best of the author's knowledge. Therefore, an effort is made in this study to investigate the effect of silica and kaolin as reinforcements in improving the thermal properties of the aluminum matrix for thermal storage. This work is aligned with the 7th, 8th, 9th, 13th, and 15th sustainable development goals of the United Nations.

2 Materials and methods

2.1 Feedstock material

The matrix material was obtained from aluminum waste. Silica sand used as reinforcement was obtained from the Silicon Research Group in Covenant University, Ota, Nigeria, and was crushed into very fine particles with a ball mill for 45 min and then sieved using a 45 μm sieve. The kaolin particle reinforcement was obtained from Apapa, Lagos, Nigeria.

2.2 Sample preparation

The matrix material was derived from aluminum waste. The waste was transferred into a steel crucible placed in

an electric resistance furnace. At 800°C, the molten aluminum was rid of the oxide layer and then decanted into a mold to solidify at ambient conditions (room temperature of 27°C and no forced convection). Table 1 presents the thermal properties of the secondary aluminum, silica and kaolin used in this study. Table 2 shows the composition of the composites. The particulates were weighed according to the compositions in Table 2 and gently poured into molten aluminum at 800°C and stirred at 500 rpm for 10 min. After taking the crucible containing the molten composite out of the furnace, the mixture was stirred for 10 s. The molten composite was poured into molds and allowed to cool down to room temperature at ambient conditions.

2.3 Sample characterization

The scanning electron microscopy (SEM) characterization was done using the Phenom ProX instrument. The samples were coated with gold and the morphology was examined. The phase analysis was carried out using XRD diffractometer (Rigaku MiniFlex 600). The analysis was done with K-Alpha radiation of 1.54060 Å using Cu as the anode. The samples were scanned from 2 to 90° at 2-theta angle with 0.02° steps.

Brinell's hardness test was carried out on the samples (SADT HARTIP 3000) to determine the influence of the silica sand particulates on the matrix. The thermal conductivity of the composites was measured (Armfield equipment model HT12) to determine the heat transfer capability

Table 2: Sample compositions

S/N	Aluminum matrix (wt%)	Silica sand (wt%)	Kaolin (wt%)
1	100	0	0
2	92.5	7.5	0
3	85	15	0
4	92.5	0	7.5
5	85	0	15

Table 1: Thermal properties of the prepared aluminum matrix and silica sand samples

	Average specific heat from 30 to 200°C ($\text{J}\cdot\text{kg}^{-1}\cdot\text{°C}^{-1}$)	Thermal conductivity ($\text{W}\cdot\text{m}^{-1}\cdot\text{°C}^{-1}$)	Density ($\text{kg}\cdot\text{m}^{-3}$)
Aluminum matrix	474.30	154.99	2649.38
SiO_2	368.95	0.06	1580.00
Kaolin	277.43	0.15	2340.00

of the composites as influenced by the reinforcements. Differential scanning calorimetry (DSC) was done (Mettler Toledo DSC822e) with a heat supply rate of $20^{\circ}\text{C}\cdot\text{min}^{-1}$ in a nitrogen atmosphere at 20 psi. The specific heat of the samples from 30 to 200°C was obtained. The density measurement of the samples was carried out using the Archimedes principle.

3 Results and discussion

3.1 Microstructural and phase analysis

Figure 1a displays the SEM/EDS of the secondary aluminum used as matrix. The top left corner of the SEM image shows the Fe-rich phase precipitated during solidification of the matrix. The presence of Fe is confirmed in the EDS graph of the aluminum sample. The diagonal lines are caused by shrinkage of the melt during nucleation, which results in pores indicated at the lower left part of the image. Alumina formed during casting is shown at the lower left part of the image. The silica sand particles shown in Figure 1b are relatively more edgy with smooth surface morphology compared to the SEM of the kaolin sample in Figure 1c, which appears to be coarser.

Figure 2(a–c) presents the SEM of the cast aluminum matrix, the cast aluminum matrix composite with 7.5 wt% silica sand and the cast aluminum matrix composite with 15 wt% silica sand, respectively. The SEM image of the secondary aluminum and the fabricated composites revealed alumina (corundum) phase was formed during and after the casting process. This is reasonably consistent with the XRD result. Fe-rich phase can be observed at the top middle region of Figure 2b. Confirmation of the presence of Fe can be observed in the EDS graph. The scaly features in Figure 2c are a consequence of shrinkage during nucleation. This results in pores as observed in the image. The diagonal lines caused by shrinkage of the composite during the cooling process is observed in the matrix presented in Figure 2c. Homogeneous distribution of the silica sand particulates is observed in the fabricated composite presented in Figure 2c. The scaly structure observed in Figure 2b is prevented in Figure 2c because it contains a higher amount of silica sand particulates which provide good enough mechanical reinforcement against shrinkage.

The composite fabricated with 92.5 wt% aluminum matrix and 7.5 wt% kaolin is presented in Figure 3b. Homogeneous distribution of the kaolin reinforcements

can be seen in the image. Fe-rich phase can be observed at the top right corner of the image. Dendrites, which are phases rich in aluminum that solidify relatively faster due to a lower melting point, can be observed in the image [5,30,31]. The dendrites serve as core onto which surrounding materials crystallize during the solidification process of the composite. In contrast, the composite fabricated with 85 wt% aluminum matrix and 15 wt% kaolin shown in Figure 3c shows the inhomogeneous distribution of kaolin particulates and a relatively rougher surface compared to the rest of the composites. Figure 4(a and b) present the cast samples' XRD results. The cast secondary aluminum sample and the composites with 7.5 and 15 wt% silica sand and kaolin, respectively, show the presence of unreacted aluminum, alumina (corundum), and kyanite. Kyanite forms from the crystallization of sillimanite, which was produced due to the reaction between alumina and silica at about 800°C .

3.2 Thermophysical properties

3.2.1 Specific heat

Table 3 presents the thermal properties of the samples. All the samples reinforced with silica sand had lower specific heat values when compared to the secondary aluminum matrix without silica sand reinforcements added. The addition of silica sand reduced the shrinkage of the composites, and this can be observed by comparing the severity of shrinkage between Figures 2b and 2c. Figure 2c had a higher amount of silica sand particulates which helped reduce shrinkage by mechanical reinforcement. Reduction in shrinkage by particulate reinforcement reduces porosity. Porous regions often serve as air traps that can store heat energy. This extra energy stored by pores can significantly increase the heat storage capacity of a material. Another similar reason for reduced specific heat can be attributed to the smooth surface of the silica sand particulates reducing the probability of interfacial air pores around them. These air pores can store heat energy with a specific heat of air higher than that of the secondary aluminum and silica sand. The higher specific heat of air is a consequence of it being a mixture of gases with the air molecules having a higher degree of freedom (or higher entropy) than solids. On the other hand, particulates with rougher surfaces with a higher probability of creating interfacial air pores will store higher amounts of heat energy in the air pores leading to a higher net specific heat of the material. The lowered specific heat of the

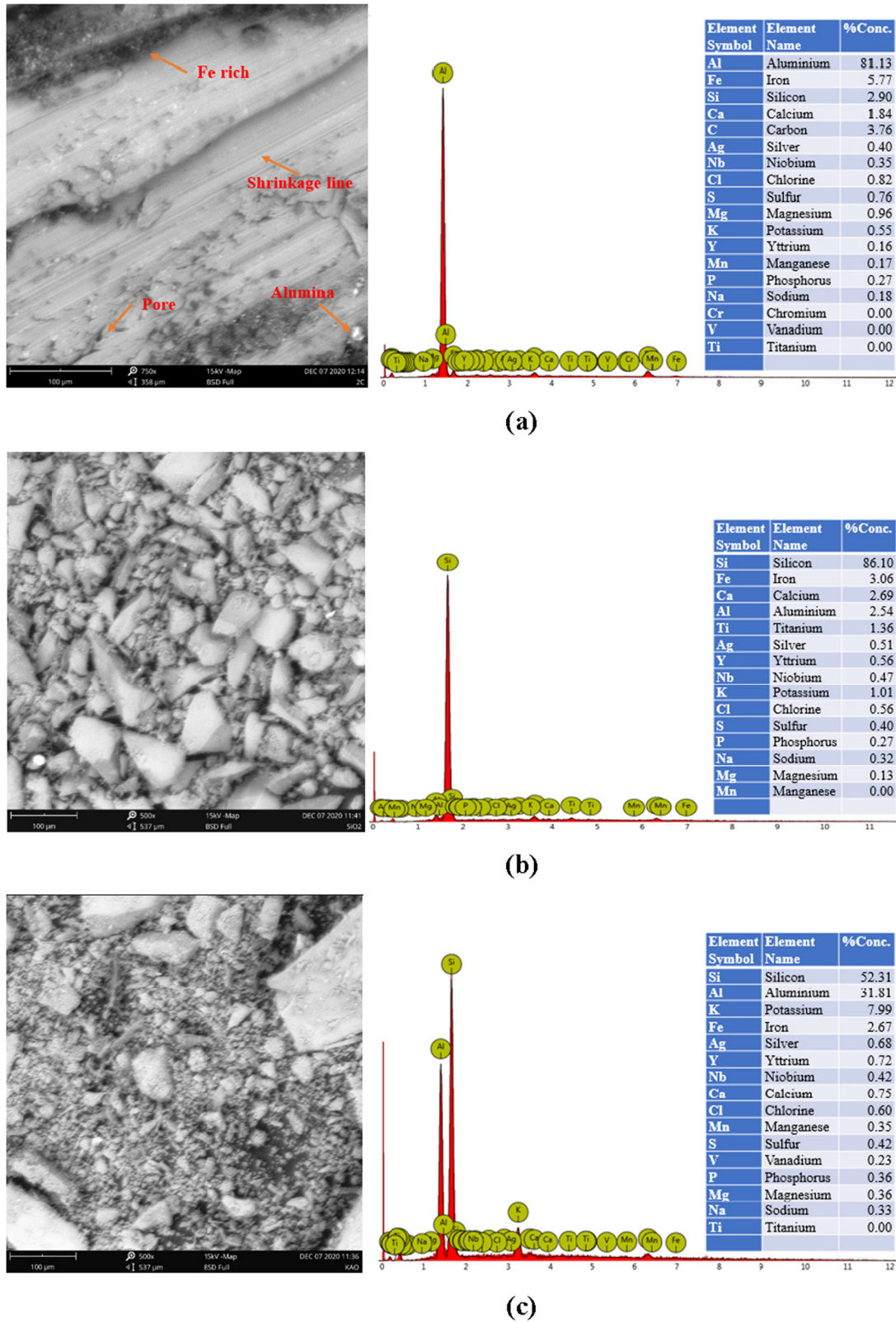
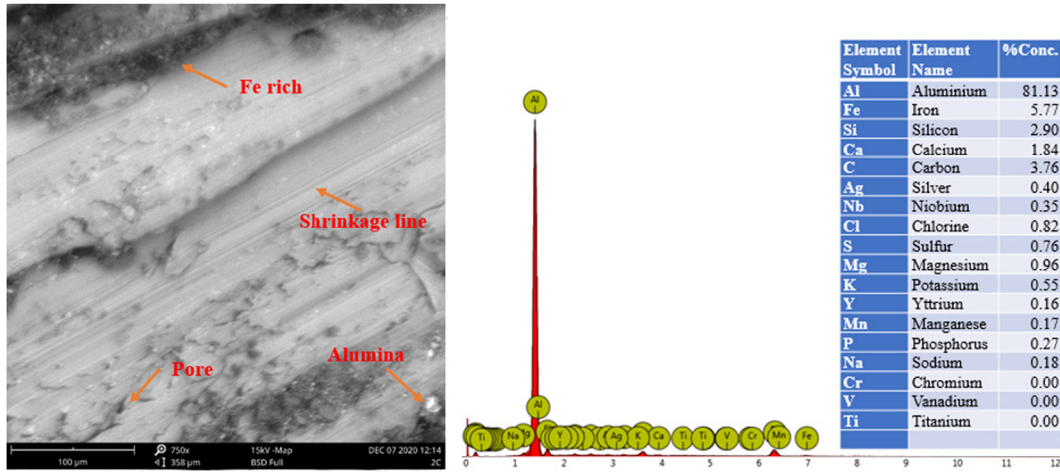
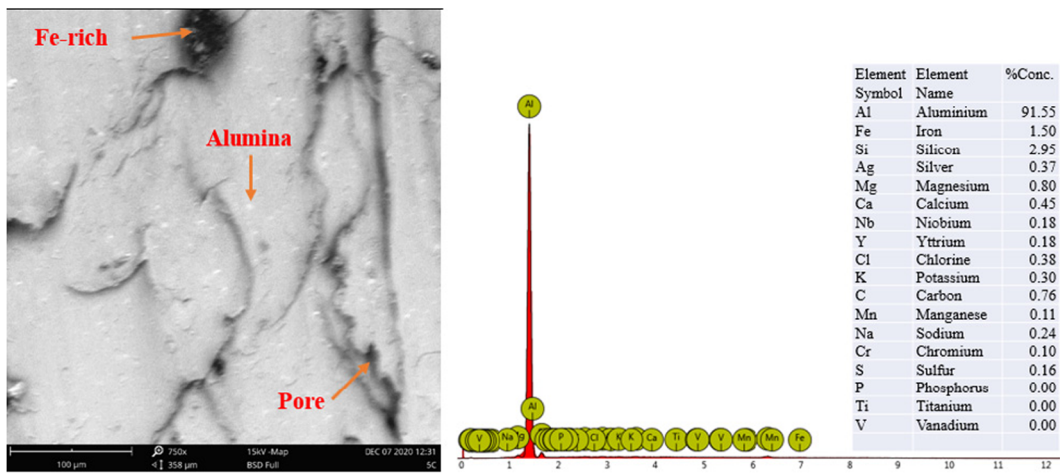


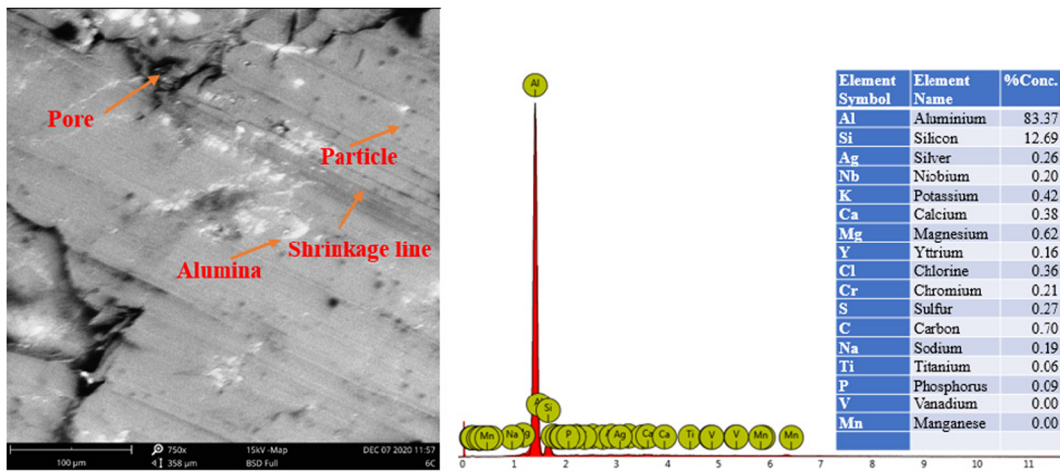
Figure 1: SEM/EDS image of (a) the aluminum matrix obtained from waste aluminum materials; (b) silica sand sample; and (c) kaolin sample.



(a)

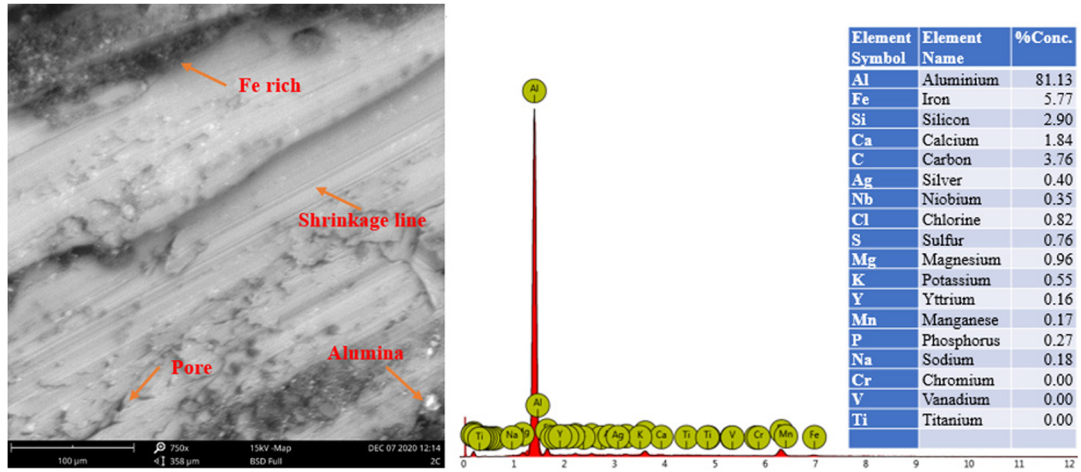


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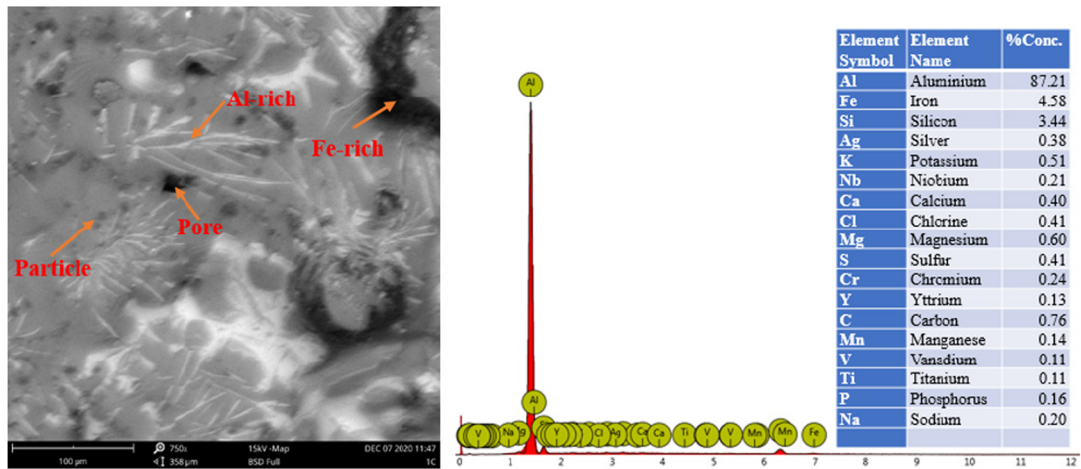


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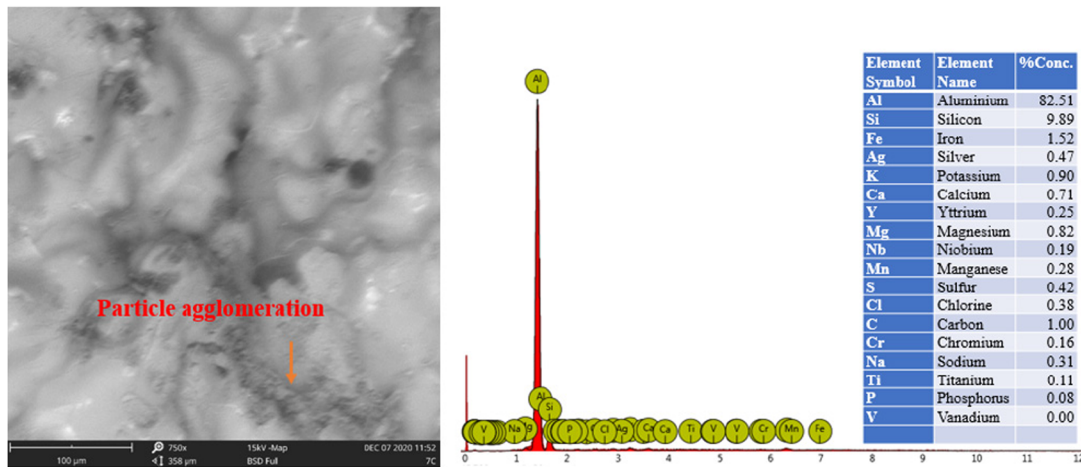
Figure 2: SEM/EDS image of (a) the aluminum matrix obtained from waste aluminum materials; (b) composite fabricated using secondary aluminum (92.5 wt%) and silica sand (7.5 wt%); and (c) composite fabricated using secondary aluminum (85 wt%) and silica sand (15 wt%).



(a)



(b)



(c)

Figure 3: SEM/EDS image of (a) the aluminum matrix obtained from waste aluminum materials; (b) composite fabricated using aluminum matrix (92.5 wt%) and kaolin (7.5 wt%); and (c) composite fabricated using aluminum matrix (85 wt%) and kaolin (15 wt%).

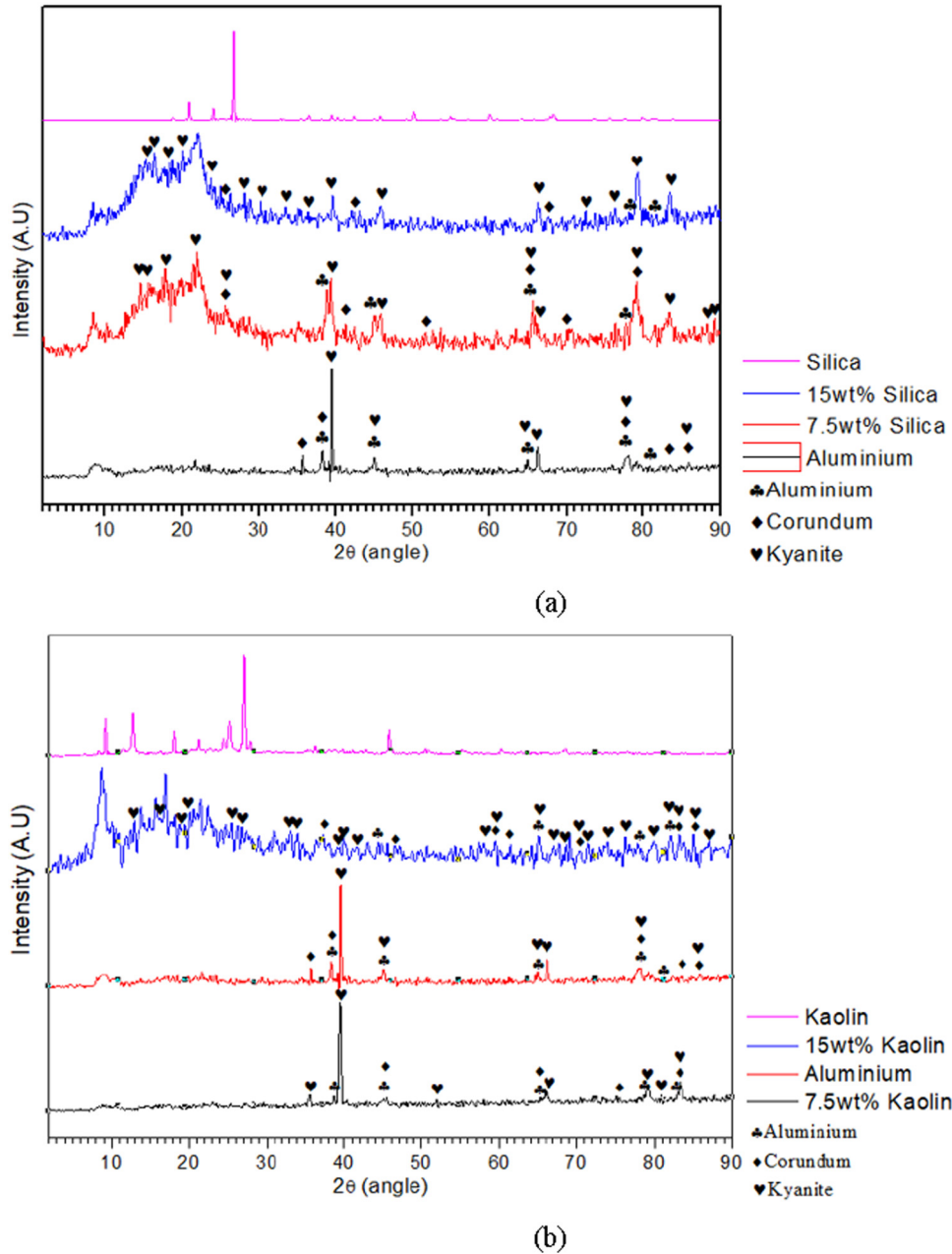


Figure 4: XRD result of (a) the silica sand sample, composites with 7.5 and 15 wt% silica sand and cast aluminum, and (b) kaolin sample, composites with 7.5 and 15 wt% kaolin and cast aluminum, showing the presence of unreacted aluminum, alumina (corundum), and kyanite.

Table 3: Prepared samples and their properties

Sample	Silica sand (wt%)	Kaolin (wt%)	Average specific heat from 30 to 200°C ($J \cdot kg^{-1} \cdot ^\circ C^{-1}$)	Thermal conductivity ($W \cdot m^{-1} \cdot ^\circ C^{-1}$)	Density ($kg \cdot m^{-3}$)	Hardness (HB)
1	0	0	474.3	154.99	2649.38	72.11
2	7.5	0	431.74	175.62	2443.47	73.11
3	15	0	382.98	181.38	2525.43	71.90
4	0	7.5	564.57	137.96	2540.83	81.38
5	0	15	679.03	148.74	2334.95	82.00

matrix as a result of adding silica sand still makes it useful as a good heat transfer material suited for applications requiring fast charging and depletion of heat energy batteries.

From Table 3, it can be observed that the increase in the amount of kaolin increases the specific heat of the matrix. Including 7.5 and 15 wt% kaolin increased the specific heat of the secondary aluminum matrix by 19% (from 474.3 to 564.57 J·kg⁻¹·°C⁻¹) and 43.2% (from 474.3 to 679.03 J·kg⁻¹·°C⁻¹), respectively. The irregularly shaped kaolin particles allow air pores at the interface between the particles and matrix. Air has higher specific heat than the matrix used, and the kaolin particulates (air is a mixture of gases with more degrees of freedom than solids) significantly add to the total heat capacity of the composite. Figure 5 presents the effect of kaolin on the specific heat of the matrix. The improved specific heat of the composites reinforced with kaolin makes them more suitable for thermal storage applications where long-term heat storage is a priority. For instance, heat energy can be stored in heat storage vehicles and transported to regions where the heat energy can be used for cooking, space heating, *etc.*

3.2.2 Thermal conductivity

For the composites fabricated with silica sand, the thermal conductivity of the composites is higher than that of the secondary aluminum matrix, as shown in Table 3. The inclusion of 7.5 and 15 wt% silica sand increased the thermal conductivity of the matrix by 13.3% (from 154.99 to 175.62 W·m⁻¹·°C⁻¹) and 17% (from 154.99 to 181.38 W·m⁻¹·°C⁻¹), respectively. Air pores act as volumes of thermal resistance

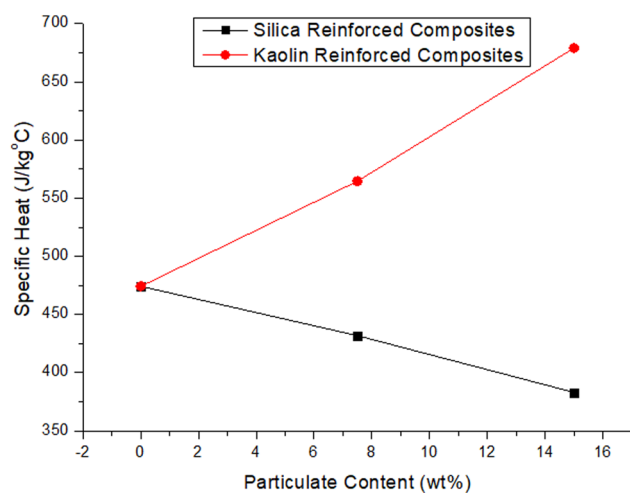


Figure 5: Specific heat of the matrix and composites reinforced with 7.5 and 15 wt% silica sand and kaolin.

[37] and, as such, limit the flow of heat by electron transfer and phonon transfer. Two other mechanisms that limit heat transfer in materials are grain boundaries and dislocations. Smooth surface silica sand particles included in the aluminum matrix will result in a composite with a low probability of interfacial pores, reducing potential volumes of space that can impede heat transfer within the material.

The addition of kaolin to the composites reduced the thermal conductivity of the matrix. This is partly attributed to the interfacial air entrapments provided by the rough morphology of kaolin particulates serving as regions of thermal resistance to heat flow [37]. Another reason is the low thermal conductivity of the kaolin particulates. These two factors combined can have a detrimental effect on the net thermal conductivity of the composites. The effect of kaolin particulates on the thermal conductivity of the matrix is displayed in Figure 6.

3.2.3 Density and hardness

The density of the matrix was reduced with the inclusion of silica sand and kaolin particulates as presented in Figure 7a. A similar reduction in matrix density with the addition of reinforcements was also observed in a previous study [32]. The stir casting fabrication method results in porous composites with gas trapped in pores. These composites will be lighter thermal storage materials than the matrix for a given volume. However, the addition of 7.5 wt% silica sand and 7.5 and 15 wt% kaolin increased the hardness value of the matrix by 1.39% (from 72.11 to 73.11 HB), 12.9% (from 72.11 to 81.38 HB), and 13.7% (from 72.11 to 82 HB) as presented in Figure 7b,

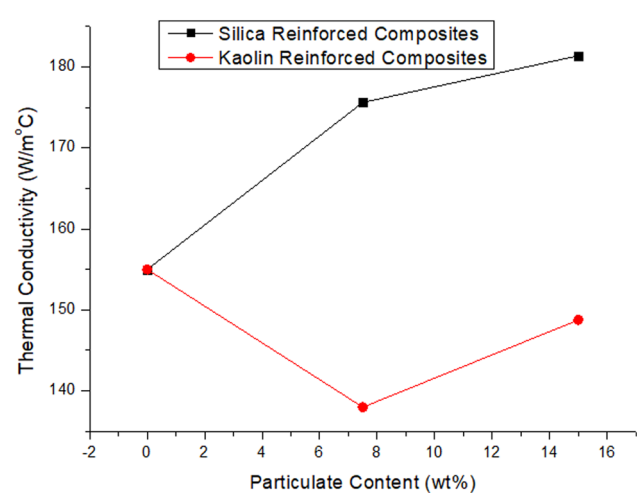


Figure 6: Thermal conductivity of the matrix and composites reinforced with 7.5 and 15 wt% silica sand and kaolin.

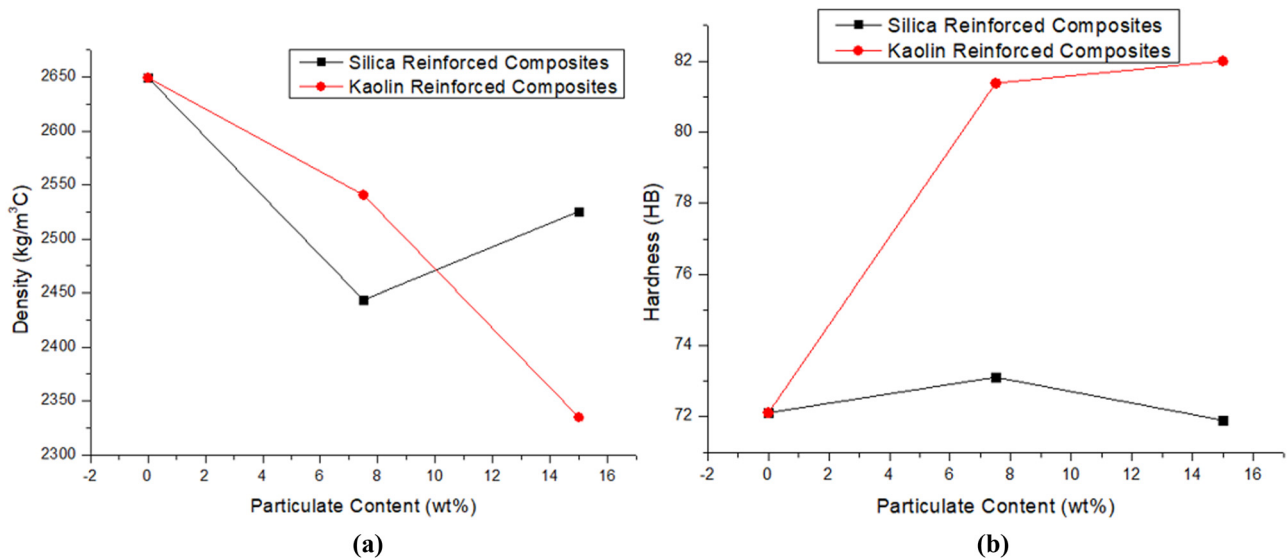


Figure 7: (a) Density and (b) hardness of the matrix and composites reinforced with 7.5 and 15 wt% silica sand and kaolin.

respectively. The increase in hardness can be attributed to the fact that metakaolin (calcined kaolin) contains more elements per molecule with strong interatomic/molecular bonds. Reinforcing aluminum matrix with kaolin will make the resulting composite denser/more compact and more resistant to deformation compared to the composite reinforced with silica sand. More elements correspond to a higher number of bonds and this results in greater resistance to deformation [38].

3.3 Heat energy stored in composites

Total heat energy stored in each composite from 30 to 200°C was determined for the compositions studied. The total heat energy is the amount of sensible energy stored by 1 kg of the materials within the specified temperature range. It can be observed from Figure 8 that the total energy stored by the materials shows a strong correlation with their average specific heat from 30 to 200°C. The

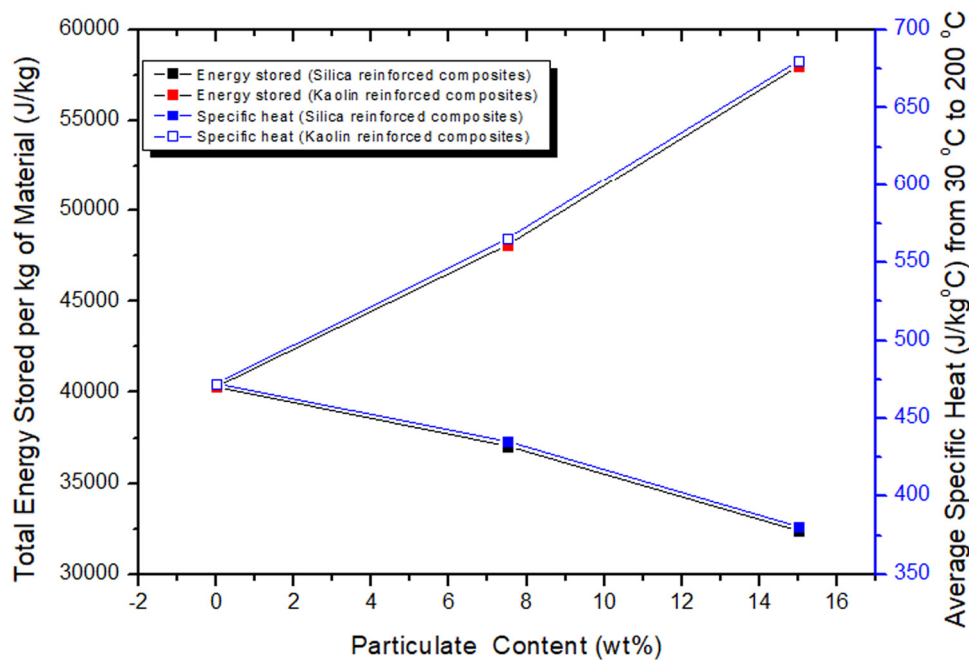


Figure 8: Total energy storage capacity in Joules·kg⁻¹ by the composites compared with the average specific heat from 30 to 200°C.

composite with the highest amount of energy stored within the temperature range is the composite reinforced with 15 wt% kaolin. From the results obtained in this work, kaolin has been found to be an ideal reinforcement with the ability to enhance the heat energy storage capacity of secondary aluminum matrix. The chemical inertness between secondary aluminum and kaolin within 30–800°C makes the resulting composites suitable for heat storage applications at a wide temperature range. Such applications include heat supply to large Stirling engines and steam turbines. The application of kaolin and silica-reinforced waste aluminum will facilitate the shift from the dependence on fossil fuels to the abundant solar energy reaching the earth. This environmentally friendly shift has the potential to significantly reduce the production of greenhouse gases and cut costs required for mining and crude oil processing activities.

The use of high-temperature thermal insulation materials studied in the work of Ndubuisi *et al.* [39] will help contain the heat energy trapped in the composite for practical applications. The advantages of aluminum as a heat storage material highlighted in Ogunrinola *et al.* [40] shows the need for further investigation on the compatibility of aluminum with thermal insulating materials at high temperatures.

4 Conclusion

The specific heat of the composites reinforced with silica sand was lower than that of the aluminum matrix. Silica sand used as reinforcement impeded the shrinkage of the material during solidification, and this reduced the formation of pores that could store heat within the material. Another reason similar to the first is the relatively smooth interfacial contact between the silica sand particulates and the matrix, leaving no space for trapped air bubbles which could increase the specific heat of the material. Compared with the aluminum matrix reinforced with silica sand, the specific heat of the aluminum matrix reinforced with kaolin increased due to the trapped air around the edges of the kaolin particulates.

The composites fabricated with silica sand particulates had higher thermal conductivity than the cast aluminum matrix. The thermal conductivity was negatively affected by interfacial pores, which acted as thermal resistance in addition to other mechanisms of thermal resistance such as grain boundaries and dislocations. The density of the matrix dropped with the inclusion of silica sand or kaolin. The hardness of the matrix was enhanced with the inclusion of kaolin reinforcements.

The improvement in hardness with the addition of kaolin is attributed to its higher molecular mass compared to that of silica sand.

Enhancement of thermal conductivity can be achieved by including silica sand for better heat energy transfer. Low-density composites will make for lightweight thermal storage materials, whereas, the specific heat of secondary aluminum has been significantly increased by the addition of kaolin, giving this type of matrix possibilities for applications in thermal energy storage in the medium to high-temperature range. Pure aluminum, which is very costly to obtain, can be replaced by a significantly cheaper alternative (secondary aluminum) and doing this is environmentally friendly.

The specific heat of the composites reinforced with kaolin and the thermal conductivities of those reinforced with silica sand and kaolin show the possibility of increase if the matrix is reinforced with these particulates beyond the 15 wt% limit in this study. Second, the effect of mixing silica sand and kaolin together to reinforce the matrix to form a hybrid composite can be explored.

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Data availability statement: The datasets generated during and/or analyzed during the current study are available from the corresponding author on reasonable request.

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